

# *Quantitative Phase Analysis using FullProf*

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# Quantitative Phase analysis with the Rietveld Method

The scale factor used in the Rietveld method is proportional to the quantity of corresponding crystalline phase

$$y_i = \sum_{\phi} S_{\phi} \left( \sum_{\mathbf{h}} \mathbf{I}_{\mathbf{h}} \Omega(T_{\mathbf{h}} - T_i) \right)_{\phi} + b_i$$
$$S_{\phi} = \frac{C}{\bar{\mu}} \frac{W_{\phi}}{(ZMV)_{\phi}}$$

D.L.Bish & S.A.Howard, J.Appl.Cryst. **21**, 86 (1988)

# Scale Factors

Experimental constant

Weight fraction of phase  $\phi$

$$S_{\phi} = \frac{C}{\bar{\mu}} \frac{W_{\phi}}{(ZMV)_{\phi}}$$

Average absorption coefficient

Unit cell volume of phase  $\phi$

Molecular weight of phase  $\phi$

Number of formula units of phase  $\phi$

The diagram illustrates the components of the scale factor equation. Red arrows point from the following text labels to the corresponding variables in the equation: 'Experimental constant' points to C; 'Weight fraction of phase φ' points to W\_φ; 'Average absorption coefficient' points to μ̄; 'Unit cell volume of phase φ' points to V in (ZMV)\_φ; 'Molecular weight of phase φ' points to M in (ZMV)\_φ; and 'Number of formula units of phase φ' points to Z in (ZMV)\_φ.

D.L.Bish & S.A.Howard, J.Appl.Cryst. **21**, 86 (1988)

# Quantitative Phase analysis with the Rietveld Method

If all phases are well crystallized one can constraint the sum of the weight fractions to 1, so that:

$$W_{\phi} = \frac{S_{\phi}(ZMV)_{\phi}}{\sum_{i=1,...n} S_i(ZMV)_i}$$

$$W_{\phi} = \frac{S_{\phi}(ZMV)_{\phi} / \tau_{\phi}}{\left[ \sum_{i=1,...n} S_i(ZMV)_i / \tau_i \right]}$$

Micro-absorption  
Brindley coefficients

D.L.Bish & S.A.Howard, J.Appl.Cryst. **21**, 86 (1988)

# Quantitative phase analysis in FullProf

Micro-absorption phenomena can be accounted through Brindley considerations:

Classification of powders according to the value of  $\mu r$  product ( $\mu$ : linear absorption coefficient;  $r$ : linear size of particles)

- . Fine powders:  $\mu r < 0.01$
- . Medium powders:  $0.01 < \mu r < 0.1$
- . Coarse powders:  $0.1 < \mu r < 1.0$
- . Very coarse powders:  $\mu r > 1.0$

# Brindley coefficients

The Brindley coefficients can be calculated iteratively by starting with the weight fractions obtained when all  $\tau = 1$  and using the empirical formula:

$$\tau_{\phi} = 1 - 1.450(\mu_{\phi} - \bar{\mu})r + 1.426[(\mu_{\phi} - \bar{\mu})r]^2$$

Expression valid for low absorption contrast  $-0.1 \leq (\mu_{\phi} - \bar{\mu})r \leq 0.1$

Mean Powders (Brindley):  $0.01 \leq 2r\mu_{\phi} \leq 0.1$

$r$  is the mean crystallite radius and  $\mu_{\phi}$  the linear absorption coefficient

G.W. Brindley, Philosophical Magazine. **36**, 347 (1945)

# Quantitative phase analysis in FullProf

$$W_{\phi} = \frac{S_{\phi}(ZMV)_{\phi} \cdot f_{\phi}^2 / \tau_{\phi}}{\sum_{i=1}^{N_{\phi}} S_i \cdot (ZMV)_i \cdot f_i^2 / \tau_i} = \frac{\textcolor{red}{S}_{\phi} \textcolor{blue}{ATZ}_{\phi} \cdot V_{\phi}}{\sum_{i=1}^{N_{\phi}} \textcolor{red}{S}_i \textcolor{blue}{ATZ}_i \cdot V_i}$$

with

$\textcolor{red}{S}_{\phi}$

Scale factor in FullProf (refinable variable)

$$\textcolor{blue}{ATZ}_i = Z_i M_i f_i^2 / \tau_i$$

FullProf parameter

$\tau_i$

Brindley factor (particle absorption contrast factor).

$\tau$  is tabulated as a function of  $(\mu_i - \mu) \cdot r$   
FullProf parameter

# Quantitative phase analysis in FullProf

$f_i$  Used to transform the site multiplicities in PCR FullProf input file, to their real values. For a stoichiometric phase,  $f = 1$  if these multiplicities are calculated by dividing the Wyckoff multiplicity  $m$  of the site by the general multiplicity  $M$  of the space group. Otherwise,  $f = occ.M/m$ , where  $occ.$  is the occupation number in the PCR file

☺ In order to GET PROPER VALUES OF WEIGHT FRACTIONS LET THE PROGRAM RE-CALCULATE **ATZ** by putting them to ZERO.

The correct **ATZ** value is rewritten in the PCR file.



# Quantitative phase analysis in FullProf

1. Crystal structure has to be refined:

**JBT=0 (IRF=0)**

→ Refine the structural parameters as usually

2. **Crystal structure is well known:**

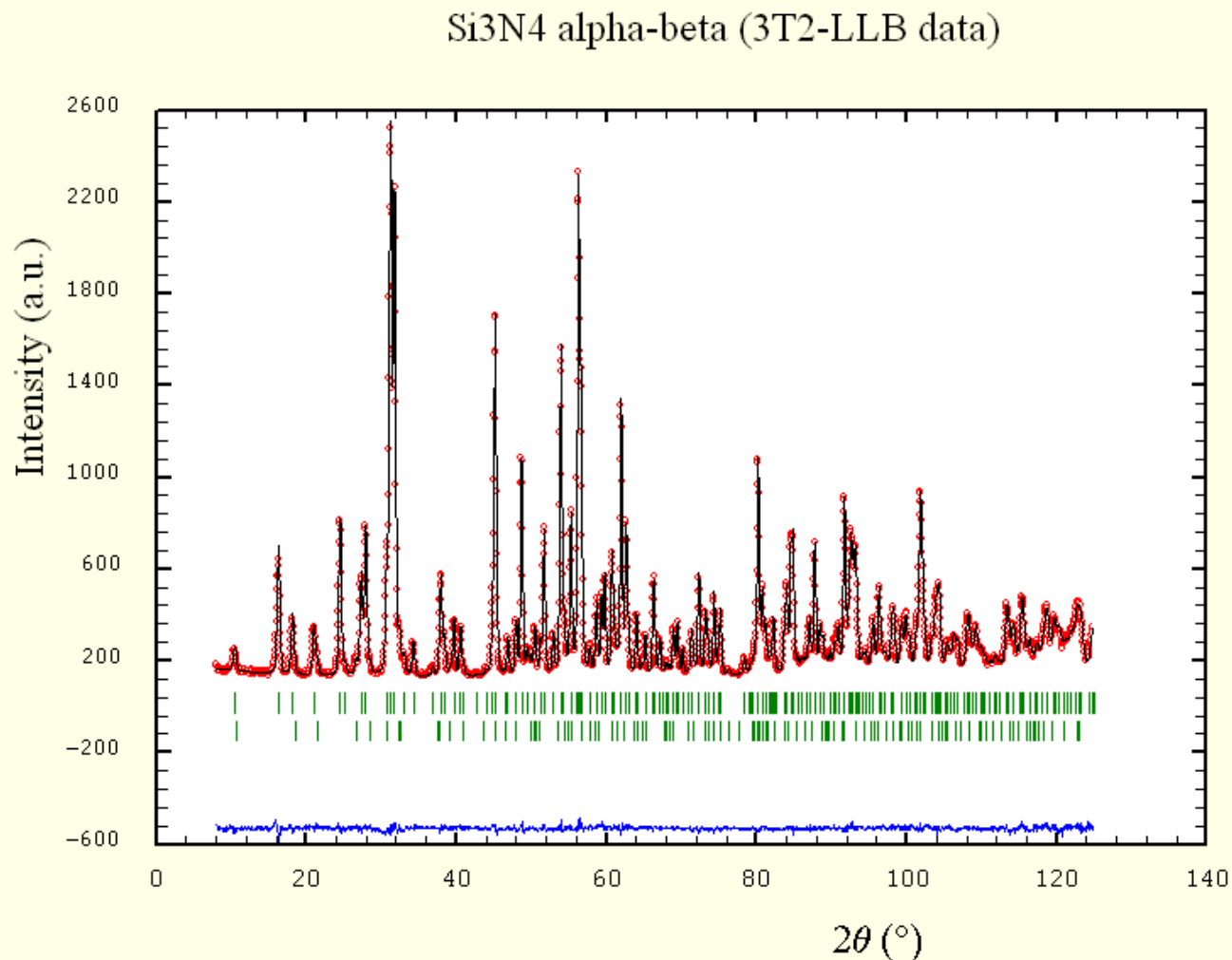
2.1 Create hkl file containing hkl list with corresponding  $F^2$   
(**JLKH=5**)

2.2 Refine the pattern without entering atomic positions  
**JBT=-3, IRF=2** (Le Bail fit mode with constant relative intensities for the current phase, but refinable scale factor)

# Rietveld Q.P.A.

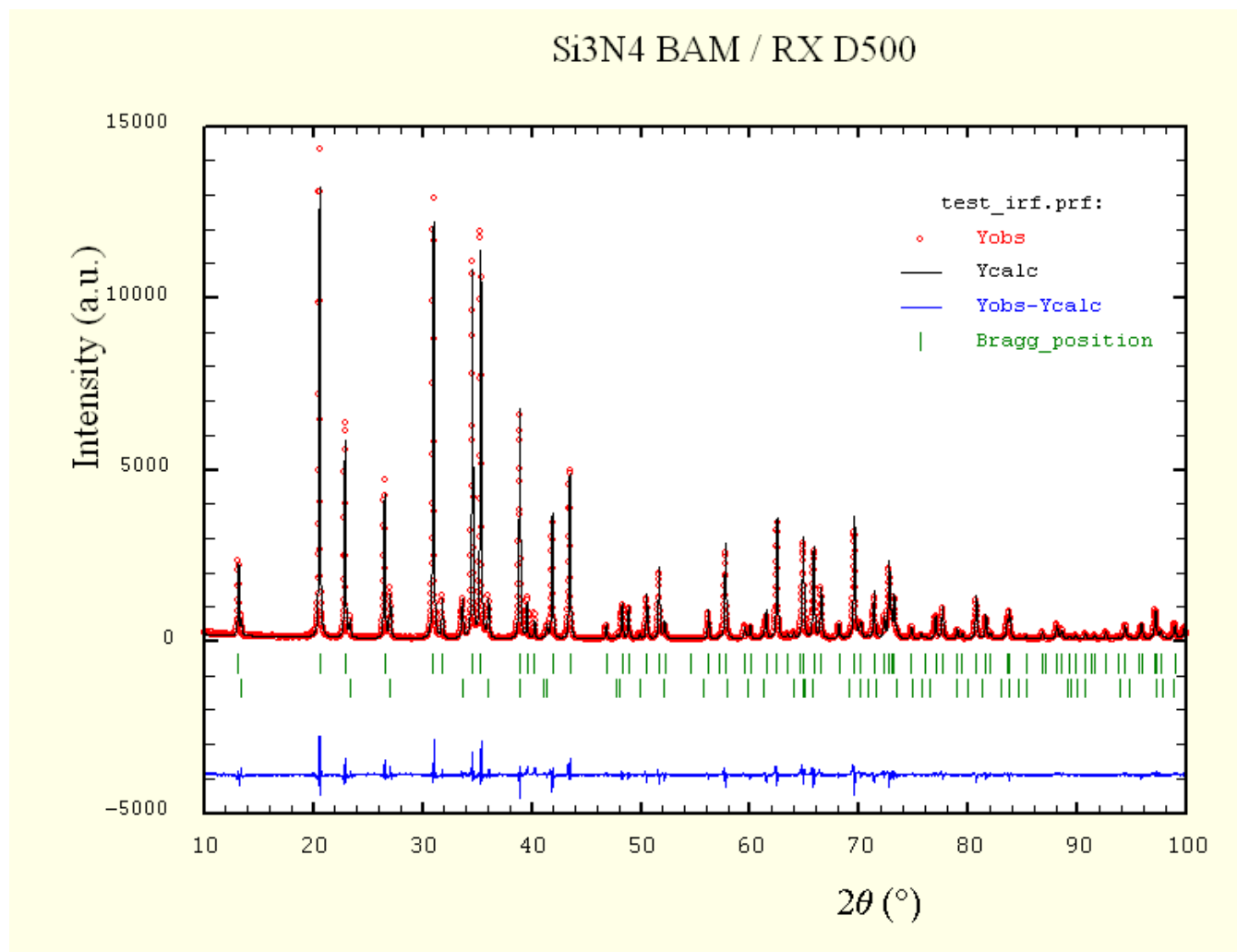
- ☺ ✓ easy to operate (automatic analysis in FullProf)
- ✓ no internal standard
- ✓ non destructive method
- ✓ up to 16 phases in FullProf
- ✓ polymorphism, microstructure
- ✓ neutron case: large amounts of powder analysis (real samples)
- ✓ industrial applications (cements, clays ...)
- ☹ ✓ structure model dependent:  $\{F_{hkl}\}$  have to be known
- ✓ beware of preferred orientation

# Rietveld Q.P.A.: $\alpha$ - $\text{Si}_3\text{N}_4$ and $\beta$ - $\text{Si}_3\text{N}_4$ mixture



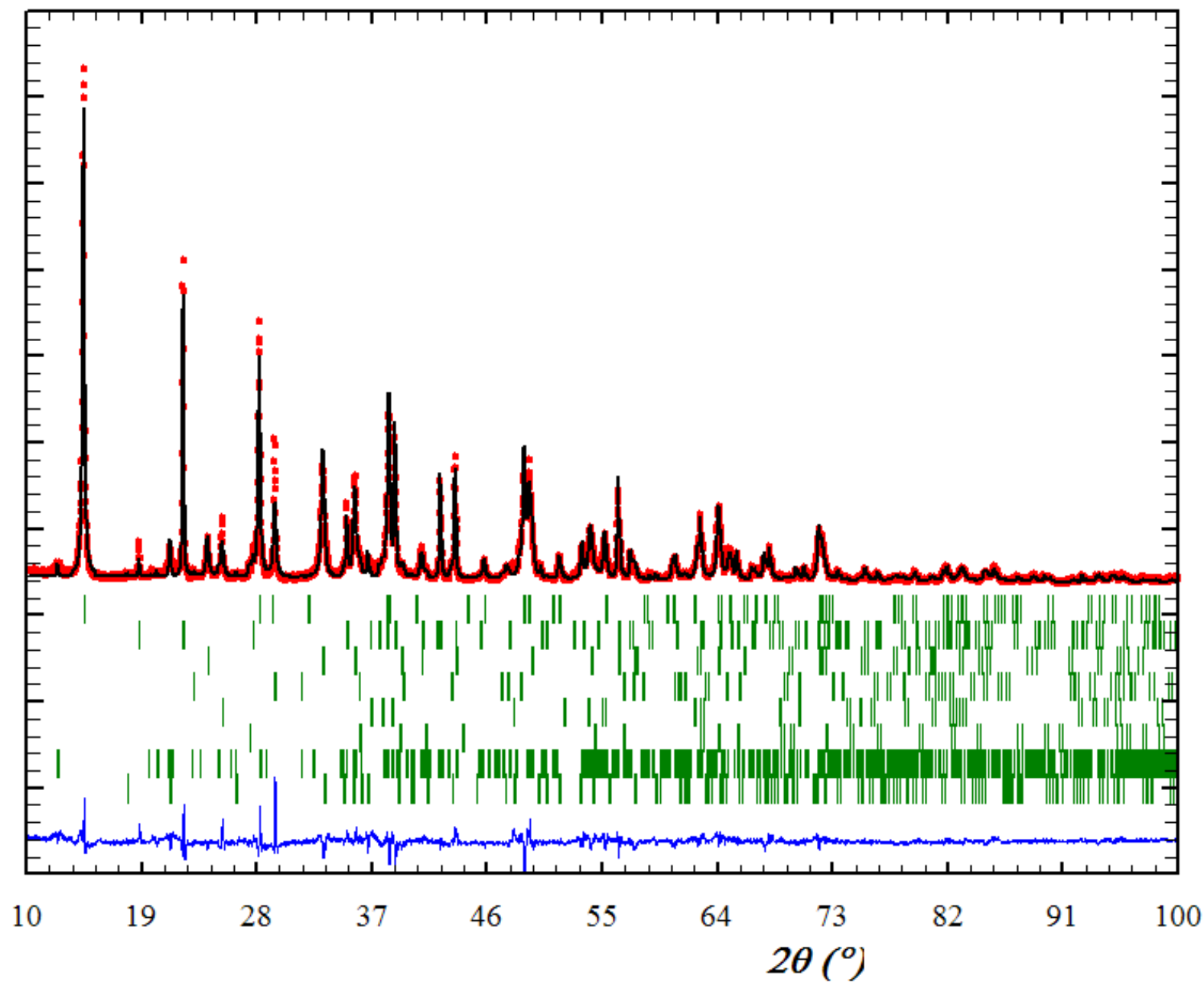
Neutron data:  $\alpha\text{Si}_3\text{N}_4$ : 93% wgt  
 $\beta\text{Si}_3\text{N}_4$ : 7% wgt

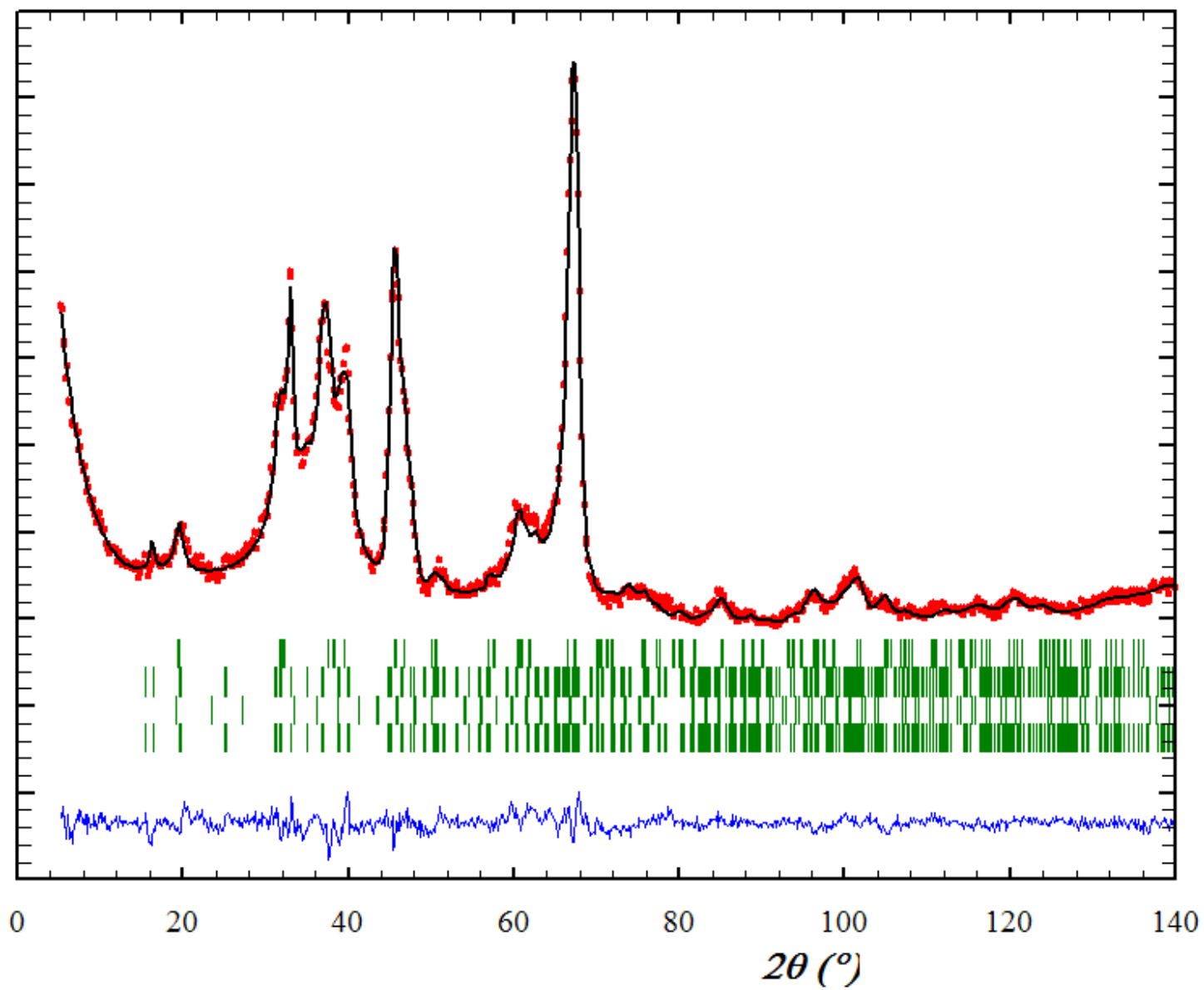
# Rietveld Q.P.A.: $\alpha$ - $\text{Si}_3\text{N}_4$ and $\beta$ - $\text{Si}_3\text{N}_4$ mixture



X-rays data:  $\alpha\text{Si}_3\text{N}_4$ : 91.6% wgt  
 $\beta\text{Si}_3\text{N}_4$ : 8.4% wgt

# Bauxite / 8 phases /Delphes Comptage long





# Some references on Q.P.A. by Rietveld method

➤ R.J. Hill & C.J. Howard, *J. Appl. Cryst.* 20, 467-476 (1987)

Quantitative phase analysis from neutron powder diffraction data using the Rietveld method

➤ G.W. Brindley, *Phil. Mag.* 36, 347-369 (1945)

The effect of grain or particle size on X-ray reflections from mixed powders and alloys considered in relation to the quantitative determination of crystalline substances by X-ray methods

➤ D.L. Bish & S.A. Howard, *J. Appl. Cryst.* 21, 86-91 (1988)

Quantitative phase analysis using the Rietveld method

➤ J.C. Taylor, *Powder Diffraction* 6, 2-9 (1991)

Computer programs for standardless quantitative analysis of minerals using the full powder diffraction profile

➤ R.J. Hill, *Powder Diffraction* 6, 74-77 (1991)

Expanded use of Rietveld method in studies of phase abundance in multiphase mixtures